Robust functionalization of amorphous cadmium sulfide films using *z*-lift amplitude modulated atomic force microscopy-assisted electrostatic nanolithography

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A robust technique, based on vertical (-lift) manipulation of a negativel biased oscillating atomic force microscope cantilever, is developed which creates raised columnar nanostructures with high aspect ratios (up to 40 nm high/150 nm wide) on amorphous CdS thin lms. The nanostructures' height (8 40 nm) is proportional to -lift of the tip and correlates with CdS lm thickness. An in-house modi ed electric force microscop is used to record the associated surface charge distribution which is found to be opposite to that of the tip. 2007 Ame ican In tit te of Ph ic . [DOI: 10.1063/1.2742910]

CdS has been widel used in photovoltaic device appli-

EFM images from the same location indicates that the charge deposited on the CdS surface remains there for 5 10 min before it nall drains via the conductive substrate. However, AFM imaging of the nanostructures 7 da s later showed no structural changes. EFM measurements indicated that the sign of the charge at the CdS surface was positive, i.e., opposite to that of the negativel biased AFM tip. It is well known that a weakl biased AFM tip forms a tin water bridge under 20% 40% ambient humidit .¹⁵ Electric breakdown in water occurs in the presence of an electric eld whose magnitude approaches $10^9 \ 10^{10}$ V m⁻¹, which is the case for the present 10 20 V biased AFM tip separated from the surface b a distance of 0.5 2 nm. It has been reported recentl ¹⁶ that electric breakdown inside the water bridge initiates eld-induced water ionization producing free carriers (electrons) as follows: H₂O=H

tric breakdown in water is common and takes place in a variet of the s stems involving high magnitude electric eld and dielectric, or in semiconductor substrates and we anticipate this to be the case in the present CdS work.

The large dc electric eld due to the biased AFM tip in pro imit to the grounded CdS lm ma e ceed 10⁸ 10⁹ V m⁻¹ because of local structural variations of the tip and CdS surface; this eld is suf cient to break Cd S bonds and trigger ionized mass transport through the CdS lms. However, electrochemical reactions at the CdS/water interface which ma also produce Cd ions cannot be ruled out. For e ample, it has been proposed that photocorrosion of CdS occurs in the presence of dissolved o gen in aque-ous solution.^{17,18} In this photocorrosion process light is essentiall an initiator which generates electron-hole pairs in the CdS lms. After a series of reactions involving the CdS lm, holes, and o gen in the aqueous solution, the net result is that SO_4^2 and Cd^{2+} ions are the main photoproducts. In the present work we propose that the large electric eld close to the biased AFM tip provides the energ required to create the electron-hole pairs in the initiation step. Then, once Cd ions have been created, o idation mechanisms and/or mass transport of the Cd ions are possible causing deposition of o ide products and/or Cd ions at the CdS surface creating the observed nanostructures. This suggests transport of material verticall from below the nanostructures towards the surface along electric eld lines. Clearl, further work is required to determine the e act nature of the mechanisms leading to nanostructure formation. A useful starting point would be the investigation of the following observations in support of a mass transport mechanism: